Scanning tunneling microscopy and kinetic Monte Carlo investigation of cesium superlattices on Ag(111)

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Cesium adsorption structures on $Ag(111)$ were characterized in a low-temperature scanning tunneling microscopy experiment. At low coverages, atomic resolution of individual Cs atoms is occasionally suppressed in regions of an otherwise hexagonally ordered adsorbate film on terraces. Close to step edges Cs atoms appear as elongated protrusions along the step edge direction. At higher coverages, Cs superstructures with atomically resolved hexagonal lattices are observed. Kinetic Monte Carlo simulations model the observed adsorbate structures on a qualitative level.

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I. INTRODUCTION

Alkali-metal adsorption on metal surfaces has been studied intensely since then (for a review the reader is referred to Refs. [1](#page-6-1) and [2](#page-6-2)). With their single s electron at the outermost atomic shell these metals are considered as simple and much research is devoted to the analysis of the interplay between their mutual interaction and the interaction with the hosting substrate.¹ Previous structural studies of $Ag(111)$ -Cs focused on adsorbate superstructures at higher coverages. For instance, a $p(2 \times 2)$ structure along with evidence of some dis-ordered and possibly incommensurate phases was found.^{3[,4](#page-6-4)} Bond length changes between the Cs adsorption layer (adlayer) and the $Ag(111)$ substrate surface were studied by surface-extended x-ray absorption fine structure also for higher coverages.⁵ Leatherman and Diehl⁶ provided a structural analysis of $Ag(111)$ -Cs for various coverages and temperatures using low-energy electron diffraction. In particular, for very low coverages they found ringlike diffraction patterns which were assigned to disordered or fluid overlayer phases. This fluid phase appears to be a common arrangement of alkali-metal films on metal surfaces at low coverage. $7-10$ It is usually argued that at low coverages the dipole-dipole repulsion between the adsorbed alkali-metal atoms dominates the alkali-substrate interaction resulting in a structure with no long-range order, yet with a typical distance between the adsorbed atoms (adatoms). Besides these structural properties, alkali-metal-covered surfaces exhibit intriguing electronic properties resulting from quantum size effects. Quantum well states confined to ultrathin films of alkali metals are investigated experimentally as well as theoretically[.11](#page-6-9)[–23](#page-6-10)

While the dipole-dipole interaction presents a direct coupling between Cs adatoms, the substrate may influence the Cs-Cs interaction in an indirect way. Indirect interaction between two atoms was first investigated theoretically by Koutecký²⁴ and then by Grimley,²⁵ Newns,²⁶ and Einstein and Schrieffer. 27 In an early field ion microscopy experiment evidence for indirect interaction between Re atoms adsorbed on $W(110)$ was reported by Tsong.²⁸ In particular, it was found that the interaction energy exhibits an oscillatory behavior as a function of the separation distance between the adatoms. Lau and $Kohn²⁹$ then predicted a long-range and oscillatory interaction between atoms mediated by Friedel oscillations of a two-dimensional electron gas. The first scanning tunneling microscopy (STM) experiment evidencing this type of interaction was reported by Brune *et al.*[30](#page-6-17) for carbon atoms adsorbed on Al(111). Similar experiments were then performed for benzene molecules on $Cu(111),³¹$ $Cu(111),³¹$ $Cu(111),³¹$ sulfur atoms on Cu(111), 32 copper atoms on Cu(111), 33 cobalt at-oms on Cu(111) and Ag(111),^{[34](#page-6-21)} and for Ce atoms adsorbed on Ag $(111).$ ^{[35](#page-6-22)} These experiments reveal that the adatomadatom distance is predominantly influenced by the surfacestate-mediated interaction. In particular, the mutual distance is reported to be $\lambda_F/2$ where λ_F denotes the Fermi wavelength of the involved surface state. This interaction may lead to superlattices on surfaces 35 or to confinement-induced adatom self-organization in quantum corrals. $36,37$ $36,37$

Here, we report a low-temperature STM experiment on Cs adsorbed on $Ag(111)$ at various coverages. At low coverages, substrate terraces are covered by Cs adatoms which locally exhibit hexagonally ordered domains. Long-range hexagonal order of the adlayer, however, does not exist owing to regions where individual Cs adatoms are not resolved in STM images. Similarly, at and close to step edges, individual Cs atoms are not resolved and appear as almost continuous rows running parallel to the step edges. At higher coverages the characteristic smearing of adsorbate structures on terraces and at step edges disappears and individually resolved adatoms form superlattices with long-range hexagonal order. Kinetic Monte Carlo simulations model the experimental situation on a qualitative level. We found that both the surface-state-mediated interaction between Cs adatoms and the dipole-dipole interaction are equally important, in particular for low coverages, and have to be included in the model. The simulations reveal that loss of atomic resolution is likely due to weakly varying particle densities in the adsorption film. While higher particle densities owing to increased dipole-dipole interactions tend to locally stabilize a hexagonal lattice, lower particle densities favor an enhanced mobility of the particles and, concomitantly, a disordered or liquidlike arrangement. At higher coverages, as a result of a reduced adatom-adatom distance, the mutual dipole repulsion is strong enough to stabilize a superlattice with longrange hexagonal symmetry and to suppress adatom diffusion.

II. EXPERIMENT

Experiments were performed using a custom-built scanning tunneling microscope operated at 7 K and in ultrahigh vacuum with a base pressure of 10^{-9} Pa. The Ag(111) surface as well as chemically etched tungsten tips were cleaned by argon-ion bombardment and annealing. Cesium was deposited at room temperature from commercial dispensers³⁸ at a rate of \approx 0.05 ML min⁻¹ as monitored by a quartz microbalance and judged from the deposition time and corresponding STM images. We define a monolayer (ML) as one Cs atom per Ag atom. All STM images presented in this work were obtained in the constant-current mode with voltages applied to the sample.

III. SIMULATIONS

We model aspects of our results by using the kinetic Monte Carlo method 39,40 39,40 39,40 which got a firm theoretical background by Fichthorn and Weinberg[.41](#page-6-27) Recently, the kinetic Monte Carlo method has been used to study the selforganization of adatoms due to surface-state-mediated interactions[.36,](#page-6-23)[37](#page-6-24) The substrate lattice is modeled by finite hop rates of adatoms between adjacent lattice sites r_i and r_j . The hop rate v_{ij} is described by an Arrhenius law according to

$$
\nu_{ij} = \nu_0 \exp(-E_{ij}/k_B T),\tag{1}
$$

where T is the substrate temperature, k_B the Boltzmann constant, and ν_0 the attempt frequency. For the hopping barrier between lattice sites *i* and *j* we use $E_{ij} = E_D + 0.5(E_j - E_i)$, where E_D is the diffusion barrier height for an isolated adatom and the $E_{j(i)}$ describe the interaction of atom *j (i)* with all the other atoms. 42 The total interaction is the sum of the dipole-dipole coupling,

$$
E_i^d = \frac{1}{4\pi\epsilon_0} \sum_{j \neq i} \frac{p^2}{|r_j - r_i|^3},\tag{2}
$$

where *p* is the dipole moment of an individual Cs adatom, ϵ_0 the dielectric constant, and the surface-state-mediated interaction, 43

$$
E_i^{\rm s} = A_0 E_0 \left[\frac{2 \sin(\delta_0)}{\pi} \right]_{j \neq i}^2 \frac{\sin(2k_F |r_j - r_i| + 2\delta_0)}{(k_F |r_j - r_i|)^2}, \quad (3)
$$

with A_0 denoting the scattering amplitude, E_0 the surfacestate binding energy, δ_0 the scattering phase shift, and k_F the Fermi wave vector.

Our idea is that regions of the adsorbate lattice that exhibit loss of atomic resolution are due to an enhanced mobility of the adatom. It is thus desirable to access the probability of finding a particle at a given lateral coordinate. Therefore, for comparison with experimental data we use the timeaveraged two-dimensional distribution function, $g_{\tau}(\mathbf{r})$, which

FIG. 1. (Color online) (a) STM image of two adjacent terraces of Ag(111) covered with 0.03-0.04 ML Cs deposited at room temperature and imaged at 7 K. A monatomically high step separates the terraces (voltage $V=200$ mV, current $I=0.2$ nA, size 97×97 nm²). (b) Close-up view of Cs adatoms on a terrace $(35 \times 35 \text{ nm}^2)$. (c) Close-up view of the adsorbate arrangement in the vicinity of a step edge $(20 \times 20 \text{ nm}^2)$.

represents the probability of finding a particle at a position **r** during a time interval τ , i.e.,

$$
g_{\tau}(\mathbf{r}) = \frac{1}{N\tau} \left\langle \sum_{i=1}^{N} \delta[\mathbf{r} - \mathbf{r}_i(t)] \right\rangle_{\tau},
$$
 (4)

where *N* is the number of particles used in the simulation.

IV. RESULTS AND DISCUSSION

A. Low coverage: *Θ* **0.1 ML**

Figure [1](#page-1-0) illustrates the main experimental findings at low coverages. The STM image in Fig. $1(a)$ $1(a)$ shows two adjacent terraces of a $Ag(111)$ surface covered with 0.03–0.04 ML Cs. On terraces we observe local hexagonal order of Cs atoms. Long-range order is suppressed in regions where Cs adatoms are no longer resolved as pointlike features [Fig. $1(b)$ $1(b)$]. Rather, these regions are characterized by broadened protrusions extending along the symmetry directions of the adsorbate lattice. From atomically resolved STM images of the adsorbate lattice a mutual adatom distance of 1.5 ± 0.2 nm was determined. At and close to step edges [Fig. $1(c)$ $1(c)$] adatoms tend to form rows parallel to the step direction separated by 1.5 ± 0.2 nm. Individual Cs adatoms start to be resolved again at distances exceeding 5 nm from the step edge.

Below we suggest that the peculiar coexistence of regions with hexagonal order together with regions in which adatoms are not resolved is likely due to locally varying Cs adatom densities. For our kinetic Monte Carlo simulations several parameters have to be estimated, which are discussed in the following. The results of the kinetic Monte Carlo simulations depend on the transition rates $[Eq. (1)]$ $[Eq. (1)]$ $[Eq. (1)]$, which are directly coupled to the parameters of the dipole and the surface-statemediated interactions. These parameters have to be estimated in advance.

In a first step we estimated the dipole moment of an individual Cs adatom. Adsorption leads to a considerable charge transfer from an alkali atom to the substrate.⁴⁴ As a consequence the adsorbed atom becomes partly ionic and develops a dipole moment which leads to a decrease in the work function. From the initial linear decrease in the work function the dipole moment of the adsorbed species may be evaluated.⁴⁴ The apparent height of the tunneling barrier in STM being related to the work function, we acquired current-distance curves on clean and Cs-covered $Ag(111)$ from which we inferred apparent barrier heights of \approx 5 and \approx 4 eV, respectively. The 1 eV difference of the apparent barrier heights is in good agreement with the work-function modifications ob-served on other surfaces covered with alkali metals.^{2[,45](#page-6-31)[,46](#page-6-32)} Using a work-function change of $\Delta \Phi \approx 1$ eV we extract a dipole moment p according to⁴⁷

$$
\epsilon_0 \Delta \Phi = e p n, \tag{5}
$$

where e is the electron charge and n the surface density of alkali-metal atoms. As a result we obtain $p \approx 0.09 \pm 0.03$ *e* nm=4.3 \pm 0.7 D.

To compare the strength of the dipole-dipole interaction $[Eq. (2)]$ $[Eq. (2)]$ $[Eq. (2)]$ with the interaction mediated by the surface state [Eq. (3) (3) (3)] we also have to determine the parameters A_0 , E_0 , and δ_0 . From previously published results^{32[–35](#page-6-22)} we infer an amplitude of $A_0 \approx 0.3$. The scattering phase shift varies between 0.3π and 0.5π (Refs. [32–](#page-6-19)[35](#page-6-22)) and within this range our simulations do not depend strongly on the specific choice of δ_0 . We therefore decided to take $\delta_0 = \pi/2$. For clean Ag(111) the surface-state binding energy is $E_0 \approx -0.06$ eV below the Fermi energy. Charge transfer from the Cs layer to the substrate most likely induces a shift of the surface-state binding energy to higher values. To estimate this shift, we resort to a similar adsorption system, namely, Cu(111)-Cs, for which a change in the $Cu(111)$ surface-state binding energy has been investigated by photoelectron spectroscopy[.48](#page-6-34) By extrapolating the photoemission results to low coverages, we find an energy shift of ≈ 0.01 eV to higher binding energies for a coverage of 0.03 ML. However, taking $E_0 \approx -0.07$ eV rather than $E_0 \approx -0.06$ eV does not alter the results of our simulations. For the parameters estimated above we calculated the interaction energies for the dipole-dipole and the surfacestate-mediated couplings according to Eqs. (2) (2) (2) and (3) (3) (3) . A comparison is shown in Fig. [2.](#page-2-0) At the adsorbate lattice spacing of $a \approx 1.5$ nm observed in the experiment (dashed line in Fig. [2](#page-2-0)) both interaction energies are similar. Therefore, in our simulations of the low-coverage adatom arrangements we take both interactions into account.

To estimate the diffusion barrier height, E_D , we observed single Cs atoms on the cold surface. Upon positioning the tip above the center of a Cs adatom the feedback loop of the instrument was opened. Next, we measured the time interval

FIG. 2. Dipole-dipole interaction energy (gray) and surfacestate-mediated interaction energy (black) between Cs adatoms on Ag(111) calculated according to Eqs. (2) (2) (2) and (3) (3) (3) with parameters $p=0.09$ *e* nm, $A_0=0.3$, $\delta_0=\pi/2$, $E_0=-0.06$ eV, and k_F $=0.813$ nm⁻¹. At the experimentally observed lattice constant of the adlayer, $a \approx 1.5$ nm (dashed line), the surface-state-mediated interaction and the dipole-dipole interaction energy have similar values.

from the moment of freezing the feedback loop until the current dropped, which signals that the adatom has moved away from its original position. This procedure resulted in a distribution of time intervals between 1 and 15 s. Together with an assumed attempt frequency of $v_0=10^{12} \text{ s}^{-1}$ (Ref. [42](#page-6-28)) we estimate a diffusion barrier height between 16 and 18 meV at 7 K. We find that the choice of $v_0=10^{12}$ s⁻¹ and $E_D \approx 17$ meV matches the time scale of the experiment quite well. However, a word of caution is necessary in this context. Simulated time scales vary with the factor ν_0 exp($-E_D / k_B T$). Consequently, by modifying ν_0 , E_D , or both within reasonable limits, experimentally observed time scales may be reproduced. If experimentally observed time scales differ from numerically simulated ones by a factor of more than $10⁴$ then it is no longer reasonable to compensate this discrepancy by an adjustment of ν_0 and E_D .^{[49](#page-6-35)} In this case new important physical processes have to be included in the menu of hopping events of the kinetic Monte Carlo algorithm. 41 It turns out that the inclusion of the adatom mutual interactions according to Eqs. (2) (2) (2) and (3) (3) (3) already leads to correlated transition processes.

With these parameters at hand we then addressed the experimentally observed coexistence of ordered and less ordered regions. Taking into account the experimental uncertainty of the Cs adatom coverage, 0.03–0.04 ML, we performed a series of kinetic Monte Carlo simulations at *T* =7 K and in a coverage regime between 0.035 and 0.040 ML. We found two possible asymptotic states: a disordered (liquidlike) state [Figs. $3(a)$ $3(a)$ and $3(c)$] and an ordered (crystalline) state [Figs. $3(b)$ $3(b)$ and $3(d)$]. Below (above) a critical coverage of $n_c = 0.0389(6)$ ML the liquid (crystalline) state is stable whereas the crystal (liquid) is metastable. Figures $3(a)$ $3(a)$ and [3](#page-3-0)(b) show density plots of $g_\tau(\mathbf{r})$ averaged over $\tau = 16$ s. In Fig. [3](#page-3-0)(a) the density plot of $g_\tau(\mathbf{r})$ exhibits regions with broad maxima, which tend to overlap and to appear as continuous lines following close-packed directions of the hexagonal particle lattice. The density plot of $g_\tau(\mathbf{r})$ shown in

FIG. 3. (Color online) $[(a)$ and $(b)]$ Density plots and $[(c)$ and (d)] defect fractions calculated by kinetic Monte Carlo simulations at $T=7$ K and $E_D=17$ meV for particle coverages 0.0383 ML in (a) and (c) and 0.0395 ML in (b) and (d). The density plots of $g_\tau(\mathbf{r})$ are calculated at 1000 s using a time averaging interval of $\tau = 16$ s. The image size corresponds to 26×26 nm². The time evolution of the defect fraction q_{τ} in (c) and (d) is shown for a disordered (triangles) and a hexagonally ordered (circles) initial particle arrangement. In the initial equilibration period between 0 and 30 s the defect fraction changes rapidly until the stable or metastable state is reached. The asymptotic stable phase is reached within 800 s.

Fig. $3(b)$ $3(b)$ is characterized by localized and sharp maxima at the sites of a hexagonal lattice with long-range order. Consequently, depending on the particle density we observe a disorder-to-order transition. In this context, the coexistence of ordered and less ordered regions of the Cs adatom lattice (Fig. [1](#page-1-0)) may be interpreted as the result of locally varying adatom coverages.

To characterize the ordered and disordered phases in more detail, we performed a Voronoï analysis.^{50,[51](#page-6-37)} This analysis provides access to local distortions of the hexagonal symmetry by determining the fraction of particles with a number of nearest neighbors deviating from 6. These particles are referred to as defects in the following. We define the defect fraction according to $q_{\tau} = N_{\tau}/(\tau N)$ with N_{τ} denoting the number of defects recorded during the time interval τ and N the total number of particles. Figures $3(c)$ $3(c)$ and $3(d)$ show the time evolution of q_{τ} for particle densities of 0.0383 and 0.0395 ML, respectively. We find that for long times $(\tau > 800 \text{ s})$ the defect fraction approaches a value above 30% (below 2%) for coverages below (above) n_c . These asymptotic defect fractions depend on the particle density but they do not depend on the initial particle configuration. In Figs. $3(c)$ $3(c)$ and $3(d)$ we show the time evolution of the defect fraction for two initial configurations. Starting from a disordered (triangles) and from a hexagonally ordered

FIG. 4. (Color online) Density plot of $g_{\tau}(\mathbf{r})$ in the vicinity of a (111) (upper part) and a (100) (lower part) step edge. The simulation area corresponds to an image size of 30×30 nm². The density plot was generated for $T=7$ K, $\tau=12$ s, and $E_D=17$ meV.

(circles) initial configuration does not alter the asymptotic defect fraction. After initialization the system is allowed to equilibrate with canonical Monte Carlo or kinetic Monte Carlo. With kinetic Monte Carlo simulations it takes about 100 s at 7 K (or \leq 100 000 kinetic Monte Carlo steps) until the system equilibrates in one of the two phases. In the vicinity of the first-order disorder-to-order transition the system can equilibrate first in the metastable phase. The stable phase emerges typically after 800 s.

To model the adsorbate arrangement at step edges, the steps are taken into account as scatterers of the surface state. We considered a step edge as an infinitely long and linear chain of Ag atoms each of which gives rise to a surface-state-mediated interaction with Cs adatoms according to Eq. ([3](#page-1-3)). Integrating the contribution of each scatterer leads to the total interaction, $43,52$ $43,52$

$$
E(r_{\perp}) = B_0 \frac{\sqrt{5}E_0}{\pi^2 k_F a_0} \frac{\sin(2k_F r_{\perp} + 2\delta_0 + \pi/4)}{\sqrt{k_F r_{\perp}^3}},
$$
(6)

where r_1 denotes the distance to the step and a_0 $= 0.287$ nm the Ag(111) lattice constant. The parameters E_0 , k_F , and δ_0 are the same that we used for modeling the adlayer structure on terraces, while the choice of $B_0=4$ meV is in good agreement with the results obtained for Cu atoms ad-sorbed on Cu(111).^{[53](#page-6-39)} Figure [4](#page-3-1) shows the density plot of $g_\tau(\mathbf{r})$ for $\tau = 12$ s. Two kinds of step edges were included in the simulations, namely, a step edge of the (111) type (upper part of Fig. [4](#page-3-1)) and a (100) step edge (lower part of Fig. 4). Deviating from our simulations of adsorbate structures on terraces we found that the surface-state-mediated interaction according to Eq. (6) (6) (6) introduced an additional stabilization of the superlattice. The distribution function, $g_{\tau}(\mathbf{r})$, exhibits overlapping maxima along the step edge directions which is in agreement with Cs adatoms showing elongated rows close to step edges in STM images [Fig. $1(c)$ $1(c)$]. Additionally, maxima of $g_{\tau}(\mathbf{r})$ become more localized again with larger distances from the step edge and reflect then the characteristic particle arrangement on terraces.

According to the Monte Carlo simulations we interpret the experimental observations as a consequence of an enhanced Cs adatom mobility at the low-coverage regime discussed in this section. We can identify the origin of the observed mobility as a property of adatom coverages, which are lower than a critical coverage leading to a disorder-to-order transition.

B. Higher coverages: *Θ* **0.1 ML**

A way to stabilize the adsorption lattice, i.e., to obtain a hexagonal superstructure with long-range order, is the increase in the coverage. Many structural analyses of alkalimetal films adsorbed on metal surfaces have previously reported the transition from a disordered phase at very low coverage to well-ordered adsorbate structures at higher cov-erages (Ref. [1](#page-6-1) and references therein). An increased coverage leads to a smaller adatom-adatom distance and will therefore increase the dipole-dipole repulsion between the adatoms. As a consequence, the probability for a Cs atom to hop from one adsorption site to an adjacent one is lowered. Our experiment and calculations corroborate this picture.

Figure $5(a)$ $5(a)$ shows an STM image of Ag(111) covered with 0.11 ML of Cs. We observe a hexagonal adlayer with long-range order and a mutual adatom distance of 0.74 ± 0.04 nm. This adsorbate superstructure corresponds to a (3×3) commensurate phase and has been reported before by low-energy electron diffraction.⁶ The inset of Fig. $5(a)$ $5(a)$ shows the atomically resolved Ag(111) lattice to indicate that the adlayer and substrate lattice exhibit the same orientation [see dashed line in Fig. $5(a)$ $5(a)$]. Regions indicating an enhanced mobility of adatoms are no longer present at this coverage. Given that the commensurate adsorbate phase is observed experimentally at a specific coverage, we performed simulations for an adatom lattice matching the substrate lattice. The superstructure similar to the one observed in the experiments was obtained as the ground state from canonical Monte Carlo calculations. The distribution function $g_\tau(\mathbf{r})$ for this coverage was then calculated for a variety of time intervals. Even for extended intervals, e.g., $\tau = 35$ s, no enhancement of the diffusion processes was observed indicating that the increased dipole-dipole coupling stabilizes the hexagonal adsorption lattice.

The same behavior was observed at higher coverages. Figure $5(b)$ $5(b)$ shows an STM image of Ag(111) covered with 0.15 ML Cs. Again, no indication of enhanced diffusion of Cs adatoms was found. The Cs adatoms are arranged in an incommensurate hexagonal superlattice with a mutual distance of ≈ 0.66 nm and rotated with respect to the Ag(111) lattice by $\approx 19^{\circ}$. Moreover, at this coverage domains of hexagonally ordered Cs atoms were observed. Cesium atomic rows of adjacent domains are offset by half an adlayer lattice constant [see inset of Fig. $5(b)$ $5(b)$]. At the domain boundaries rows of irregularly shaped structures occur whose apparent height depends on the tunneling voltage. The formation of domains may be understood in terms of stress release at the domain boundaries. Similar observations of translational adsorbate domains have been reported for oxygen adsorption on W(110) (Ref. [54](#page-6-40)) and on Rh(111).^{[55](#page-6-41)}

FIG. 5. (Color online) (a) Quasi-three-dimensional representation of constant-current STM image of $Ag(111)$ covered with 0.11 ML of Cs ($V=0.25$ V, $I=0.1$ nA, 10×10 nm²). Cesium adatoms which appear as almost circular protrusions exhibit a mutual distance of 0.74 nm. Inset: atomically resolved Ag (111) lattice. The dashed line indicates that adlayer and substrate lattice have the same orientation. (b) STM image of Cs-covered Ag (111) at 0.15 ML $(V=1.2 \text{ V}, I=0.2 \text{ nA}, 80.4 \times 80.4 \text{ nm}^2)$. Lines on terraces which contain irregularly shaped structures appearing as depressions at the applied tunneling voltage are boundaries between translational domains of the adsorbate lattice. Inset: Atomically resolved Cs domains. While the orientation of the adjacent domains is identical, the lattices are translated by half a superlattice constant (see dashed line).

To model these observations within a kinetic Monte Carlo approach we simulated particle arrangements for coverages ranging between 0.11 and 0.2 ML (Fig. [6](#page-5-0)). After an interval of $\tau = 12$ s is elapsed the coordinates of 340 particles with a coverage of 0.11 ML are given by the upper left plot in Fig. [6.](#page-5-0) This coverage corresponds to the (3×3) commensurate adsorption phase and we see that the superlattice aligns with the crystallographic direction of the substrate lattice depicted as the horizontal arrow). With increasing coverage we observe that this alignment weakens. At 0.132 ML some domains are still oriented along the substrate crystallographic direction while others enclose an angle of $\approx 20^{\circ}$ indicating that at 0.132 ML commensurate and incommensurate phases coexist. This coexistence is in agreement with previous re-

FIG. 6. (Color online) Arrangement of 340 particles according to kinetic Monte Carlo simulations using a time averaging interval of $\tau = 12$ s and a substrate temperature of $T = 7$ K. The coverages are indicated at the top right of each plot. The horizontal (inclined) arrow indicates the crystallographic direction of the substrate (particle lattice). The rotation angle is defined as the smaller angle enclosed by the two arrows.

ports based on low-energy electron diffraction.⁶ At 0.136 ML all particle domains exhibit a rotation angle of $\approx 20^{\circ}$. This angle is maintained until at 0.2 ML the particle lattice is aligned with the substrate lattice again. In particular, the incommensurate phase observed experimentally at 0.15 ML with a rotation angle of $\approx 19^{\circ}$ is well reproduced by our simulations. We summarize calculated and measured rotation angles as a function of the coverage in Fig. [7.](#page-5-1) We notice that our simulations reproduce the experimentally observed rotation angles for commensurate adsorbate phases. The continuous rotation of incommensurate phases reported for Ag(111)-Cs (Ref. [6](#page-6-6)) and Cu(111)-Cs,¹⁰ however, is not concerned here.

Besides the rotation angle of commensurate superstructures the simulations indicate the origin for domain formation as observed experimentally at 0.15 0.15 ML [Fig. $5(b)$]. Increasing the particle density in the simulations to 0.14 ML and thus being close to the $(\sqrt{7} \times \sqrt{7})R19.1^\circ$ commensurate phase leads to a particle arrangement where the domains

FIG. 7. (Color online) Rotation angle of the superlattice with respect to the substrate lattice. Experimental data are depicted as circles and calculated data as triangles. The dashed line indicates a coverage at which two phases with different rotation angles coexist.

have almost disappeared in favor of a nearly homogeneous particle film with long-range hexagonal order. At a particle coverage of 0.155 ML the boundaries of adjacent domains overlap leading to an increase in the particle interaction at the boundaries. A further increase in the particle density to 0.2 ML induces a decrease in the domain size and leads to a similar particle arrangement as observed for 0.136 ML. As a result we obtain that particle coverages giving rise to incommensurate phases exhibit the propensity to form domains, while commensurate phases lead to homogeneous films.

V. SUMMARY

Cesium films adsorbed on $Ag(111)$ with submonolayer coverage have been investigated by scanning tunneling microscopy. The resulting adsorbate arrangements are modeled by kinetic Monte Carlo simulations. At very low coverages, long-range hexagonal order of the superstructure does not exist while at higher coverages a stable and hexagonally ordered adsorbate lattice is observed. Modeling of the adsorbate arrangement at very low coverages demonstrates the importance of both dipole-dipole repulsion and surface-statemediated interaction between Cs adatoms. Long-range hexagonal order is inhibited if the coverages are lower than critical coverages leading to a disorder-to-order transition. At higher coverages, the dipole-dipole repulsion between the Cs adatoms becomes large enough to stabilize hexagonal superlattices. While commensurate adsorption phases are characterized by a homogeneous adsorbate film, incommensurate phases exhibit domainlike patterns. The experimentally observed rotation of commensurate adsorbate lattices is reproduced by our calculations.

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